

# A stochastic tessellation for modelling and simulating colour aluminium grain images

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## Summary

The work described in this article is motivated by the need for modelling and simulating some colour aluminium grain images. One possible approach for modelling these colour aluminium grain images is to approximate the images by trivariate piecewise constant functions corrupted by additive and correlated noise. Such a piecewise constant function modelling approach is naturally divided into three parts: shape, colour and noise. The main focus of this article is to present a new tessellation model which can be applied to model the shape characteristics of the aluminium grain images. One special feature of the new tessellation model is that it is capable of generating tessellation patterns having rough boundaries. Some empirical properties and variants of the tessellation model are also described. Then, as a secondary focus, this article describes models and methods for modelling the observable colour values and noise of the aluminium grain images. Both visual and empirical comparisons between real and simulated images suggest that our modelling is reasonable.

## 1. Introduction

Most of the material to be presented in this article is motivated by the need for modelling highly complex tessellated images such as those displayed in Fig. 1. These images show grains in rolled aluminium coils, viewed through polarized light. These images are two of a series provided to the CSIRO Division of Mathematics and Statistics by Comalco Aluminium Ltd, Sydney, several years ago. Note the elongated nature of many of the grains. This is due to the fact that the coils have been rolled (i.e. flattened with a roller). As part of its quality control

procedures, Comalco was interested in obtaining an understanding of the variability within and between coils of quantities such as grain area, perimeter and orientation.

The first step to extract such quantitative information is to segment the images into their constituent grains. For example, see Fig. 2 for a semi-automatically obtained segmentation of the image displayed in Fig. 1(a). In order to evaluate and compare quantitatively the performance of different segmentation procedures, it is useful to develop models and methods which can be used to simulate the important characteristics of these aluminium grain images.

A close examination of the images suggests that a colour aluminium grain image can be well approximated by a trivariate two-dimensional (2D) piecewise constant function with noise, possibly correlated, superimposed. That is, grains and grain boundaries can be represented respectively by connected regions and discontinuity points of a trivariate 2D piecewise constant function. Such a piecewise constant function model can be simulated, in a natural manner, by the following three steps:

1. Shape: Generate a spatial *tessellation*. Each region of the generated tessellation will be treated as a single aluminium grain, and boundaries of the tessellated regions will be treated as grain boundaries.
2. Colour: Given a tessellation, assign a *simulated mean colour value* to each of the generated 'empty' regions. Such simulated mean colour values should resemble the observed mean colour values of the real aluminium grain images.
3. Noise: Given a 'colour-assigned' tessellation, add suitable (possibly *correlated*) noise to it. Such noise, from a visual sense, should corrupt a noise-free 'colour-assigned' tessellation in a similar manner to the noise in the real aluminium grain images.

The main contribution of this article is the proposal of a new tessellation model which can be applied to model the

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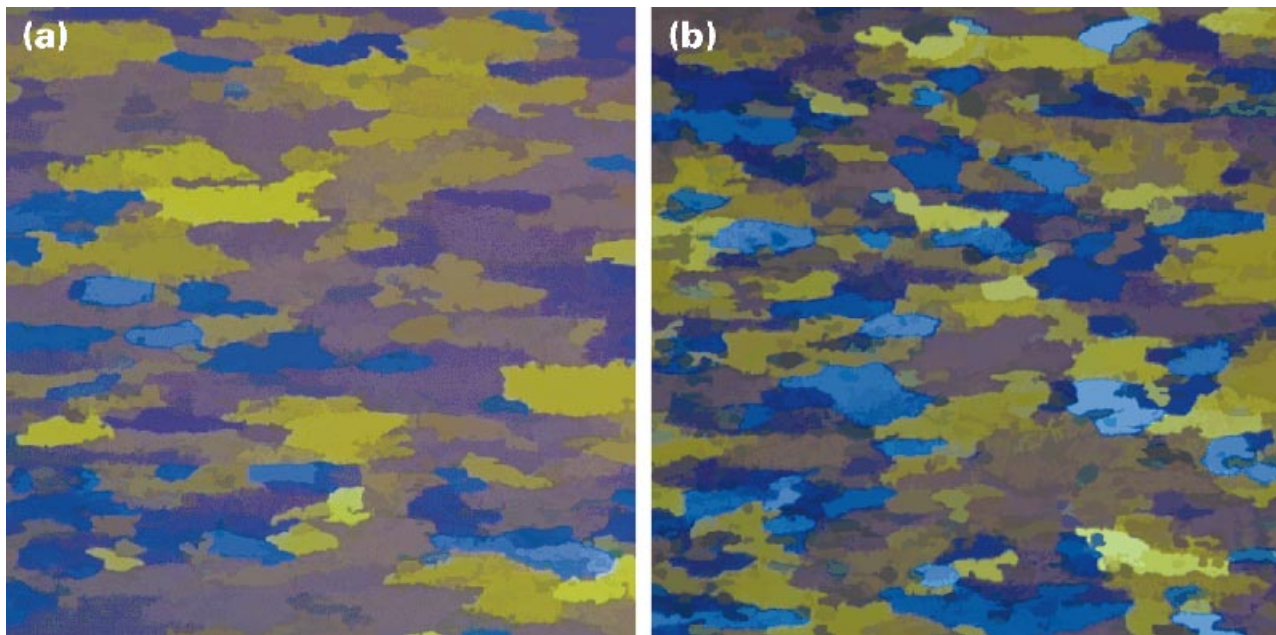


Fig. 1. Two real aluminium grain images.

shape characteristics of the aluminium grain images (part of the material has been presented at an earlier conference; see Lee & Cowan, 1994). In particular, this new model incorporates quite naturally the rough boundaries that are clearly visible in the aluminium grain images. In addition, we also describe methods for modelling and simulating the colour and noise characteristics of the aluminium grain images.

The rest of this article is organized as follows. Section 2 presents the new tessellation model. One key component of the model is a *stochastic growth mechanism*, and Section 3 compares this growth mechanism with other growth mechanisms appearing in the literature. Some properties of the tessellation model are presented in Section 4, while some possible variants of the tessellation model are described in Section 5. Section 6 demonstrates how the tessellation model can be applied to simulate the shape characteristics of the aluminium grain images. Sections 7 and 8 describe, respectively, methods for modelling the colour and noise characteristics of the aluminium grain images. Then, in Section 9, real and fully simulated images are empirically compared. Finally, Section 10 concludes this article.

## 2. The proposed tessellation model

This section presents our tessellation model. We mention again that the real grain boundaries are visually rough, and that the proposed tessellation model is capable of generating tessellation patterns with rough boundaries.

### 2.1. Why are existing tessellation models not appropriate?

Many common tessellation models have been proposed in the literature for the modelling of image data. However, we believe that they are not as appropriate as our proposed model. First, many of the common models, such as the Voronoi, Delaunay and Poisson line tessellations, are

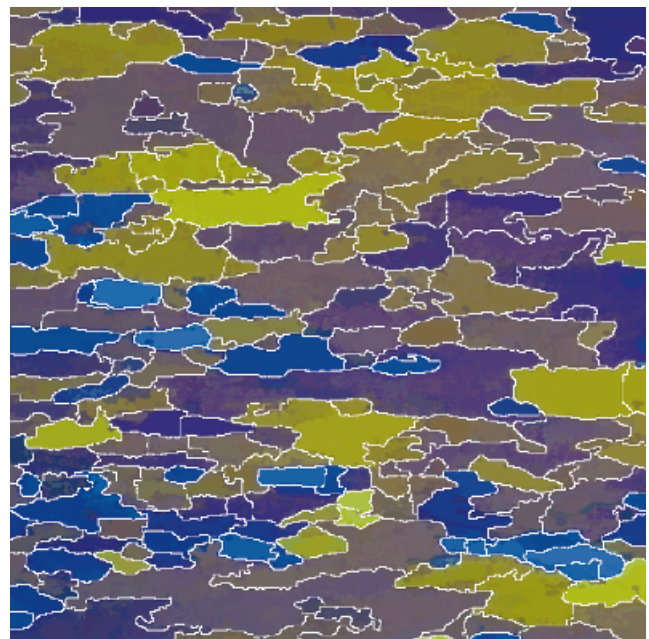


Fig. 2. A semi-automatic segmentation of the image displayed in Fig. 1(a).

inappropriate as their boundaries consist of straight lines (i.e. visually smooth); see, for example, Ahuja & Schachter (1983), Stoyan *et al.* (1987), Okabe *et al.* (1992), Schwertel & Stamm (1997) and references given therein.

There is one class of models, namely coverage models, which can be applied to generate tessellations with rough boundaries. These tessellation models are also known as 'bombing' models (e.g. Ahuja & Schachter, 1983), dead-leaves models (e.g. Matheron, 1968, Jeulin, 1979, Jeulin, 1980, 1993) and falling-leaves tessellations (e.g. Cowan & Tsang, 1994). The idea is that flat objects are successively dropped onto an image, with those dropped later covering those dropped earlier. Then, when the process finishes, visible object boundaries form a tessellation. Thus, a tessellation with rough boundaries can be obtained if the boundaries of those flat objects are rough. However, this object dropping mechanism bears no relation to the physics (see below) of the formation of the real aluminium grains, so we regard it as not as appropriate as our proposed model, which does follow the physics to some extent.

## 2.2. Seed-and-grow strategy

It is known that the formation of those aluminium grains follows some sort of growth process. Therefore it is natural to build our tessellation using a 'seed-and-grow' strategy. That is, we start with a number of seeds (or nuclei, or centres) and grow grains outward from the seeds. Eventually the grains meet each other and those points of contact define grain boundaries, which define a tessellation of the space. The Voronoi tessellation is one example of this: all seeds grow with uniform growth rate in all directions simultaneously, and boundaries are formed when two growing regions meet. As a result of the uniform growth rate, these boundaries are smooth. In order to produce rough boundaries, the proposed tessellation employs a growth mechanism having a *stochastic growth rate*.

Besides producing rough boundaries, another major characteristic of the proposed tessellation is that it is defined on a  $d$ -dimensional regular grid (e.g. square, triangular or hexagonal) rather than the traditional  $\mathbb{R}^d$ , for the purpose of facilitating computer simulation. We first restrict ourselves to the case of  $d=2$  and the grid is square with a four-connected neighbour system. Extensions to higher dimensions and other regular grids are straightforward; see Section 5.2.

## 2.3. Initial pixel allocation

The tessellation building process is divided into two stages: *initial pixel (or seed) allocation* and *pixel-labelling (or seed-growing)*. The second pixel-labelling stage can be further divided into two components: *pixel choosing* and *pixel*

*assigning*. In this article a digital version of a two-dimensional homogeneous Poisson process is used to perform the initial pixel allocation. We could certainly use other spatial point processes for this initialization, but we will consider only the Poisson case here.

## 2.4. Pixel-labelling

Suppose  $n$  initial pixels are generated (and hence there are  $n$  grains in the final realization) and we group these  $n$  pixels into  $n$  sets:  $G_1, G_2, \dots, G_n$ . Thus initially each  $G_i$  has exactly one pixel. The pixel labelling mechanism labels one pixel at a time according to the stochastic rule described below, and the whole process stops when all pixels have been labelled.

At any moment during the labelling process, the domain of interest (i.e. the image to be tessellated) is partitioned into three sets  $L$ ,  $C$  and  $N$ , and every pixel of the image belongs to exactly one of these three sets.  $L$ ,  $C$  and  $N$  represent 'Labelled', 'Candidates for labelling' and 'Not candidates for labelling' respectively, and they are defined as:

$$L = \bigcup_{i=1}^n G_i,$$

$$C = \{x \in L \mid nbr(x) \cap L \neq \emptyset\},$$

$$N = \{x \in L \cup C\},$$

where  $nbr(x)$  denotes the set of all four-connected neighbours of the pixel  $x$ . Hence before the first labelling,  $L$  is the set of all initial pixels. With this setting, labelling a pixel is equivalent to *choosing* an  $x \in C$  and *assigning* it to exactly one of the  $G_i$ s. Of course, after this pixel assignment we have to remove  $x$  from  $C$  and perform necessary updates on all related sets.

The stochastic labelling mechanism is as follows. First we further partition  $C$  into four sets  $C_1, C_2, C_3$  and  $C_4$ , and define  $n_1, n_2, n_3$  and  $n_4$  as:

$$C_i = \{x \in C \mid \text{number of labelled neighbours of } x \text{ is } i\},$$

$$n_i = \text{number of elements in } C_i,$$

for  $i = 1, \dots, 4$ . We also associate each  $C_i$  with a tuning parameter  $\lambda_i$ . Now, for any  $x \in C_i$  and conditional on  $L$ , we assume

$$P(x \in C_i \text{ is the next pixel for labelling} \mid L) = \frac{\lambda_i}{\sum_{j=1}^4 n_j \lambda_j}, \quad (1)$$

where  $P(E)$  is the probability of the event  $E$ . Equation (1) merely says that the probability that an  $x \in C_i$  will be chosen as the next pixel for labelling is proportional to  $\lambda_i$ . If  $\lambda_i = i$  for  $i = 1, \dots, 4$ , it translates to: the probability that a particular pixel will be chosen as the next labelling pixel is proportional to how many labelled neighbours that this pixel has. Note that Eq. (1) is the key equation which drives

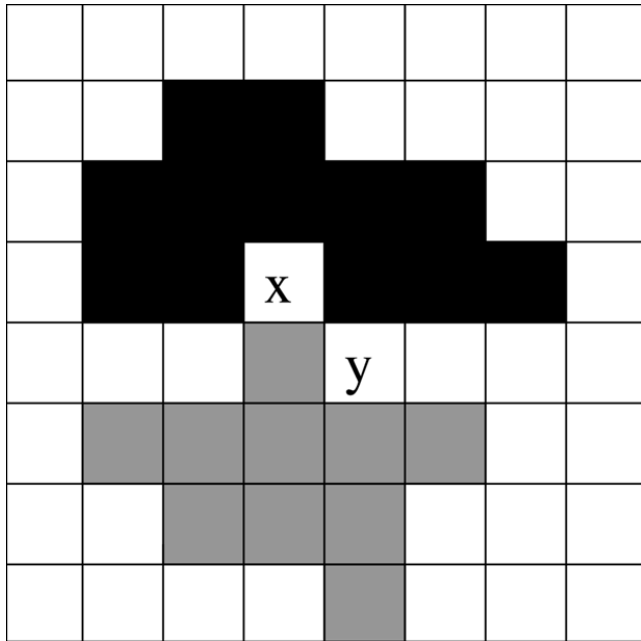


Fig. 3. White squares are unlabelled pixels. Using the random rule, pixels  $x$  and  $y$  will be labelled black with probabilities  $3/4$  and  $1/3$  respectively. Using the majority rule,  $x$  will be labelled black while  $y$  will be labelled grey.

our stochastic growth mechanism, and that those  $\lambda_i$ s are parameters of our tessellation.

After having chosen the next pixel  $x$ , say, for labelling from the  $C_i$ s using Eq. (1), we have to assign this chosen pixel  $x$  to exactly one of the  $G_i$ s. If all labelled neighbours of  $x$  belong to the same  $G_i$ , then  $x$  will be assigned to this same  $G_i$ . If those labelled neighbours of  $x$  come from more than one  $G_i$  (in this case  $x$  is a boundary pixel), then we could assign  $x$  to  $G_i$  with probability proportional to the number of neighbours of  $x$  already assigned to this  $G_i$ . We call this boundary formation rule the 'random rule'. We could also have a 'majority rule': assign  $x$  to the set  $G_i$  which contains most of those labelled neighbours of  $x$ . See Fig. 3 for an illustrative example.

Repeat this pixel choosing and assigning mechanism until all pixels have been labelled (or equivalently, when  $C$  and  $N$  both become empty). This completes the algorithm but a further step of rescaling the vertical axis of such tessellations with respect to the horizontal axis helps to produce simulated images which appear to be similar to images of rolled aluminium grains. Figs 4 and 5 provide realizations of the proposed tessellation and the growth process respectively.

We close this section with the following remark about those grains that intersect the image boundaries. The shapes of these grains are 'biased' in the sense that, during the growth process, no grains with seeds outside the image are growing towards the centre of the image and interact

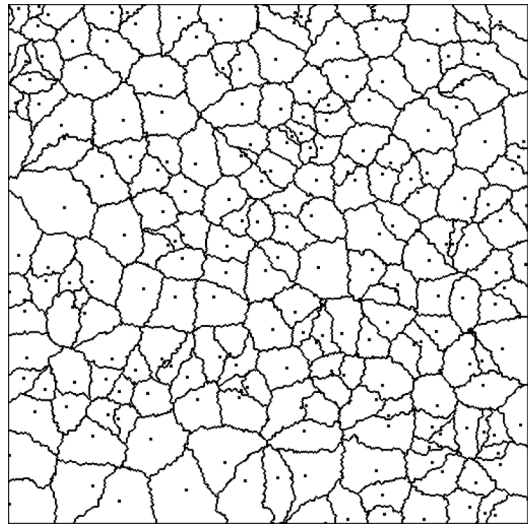


Fig. 4. A realization of the proposed tessellation with Poisson intensity  $0.0008$ ,  $r_1 = r_2 = 1$  ( $r_1, r_2$  are defined in Section 4.1) and the random rule for boundary formation. Dots inside grains are initial seeds generated by the Poisson process. Since in our formulation boundaries are composed of horizontal and vertical 'edges' between pixels, the actual boundaries cannot be displayed exactly in images. The way we deal with this is to display all pixels which have at least one of their four edges as part of a boundary. Size of image:  $512 \times 512$ .

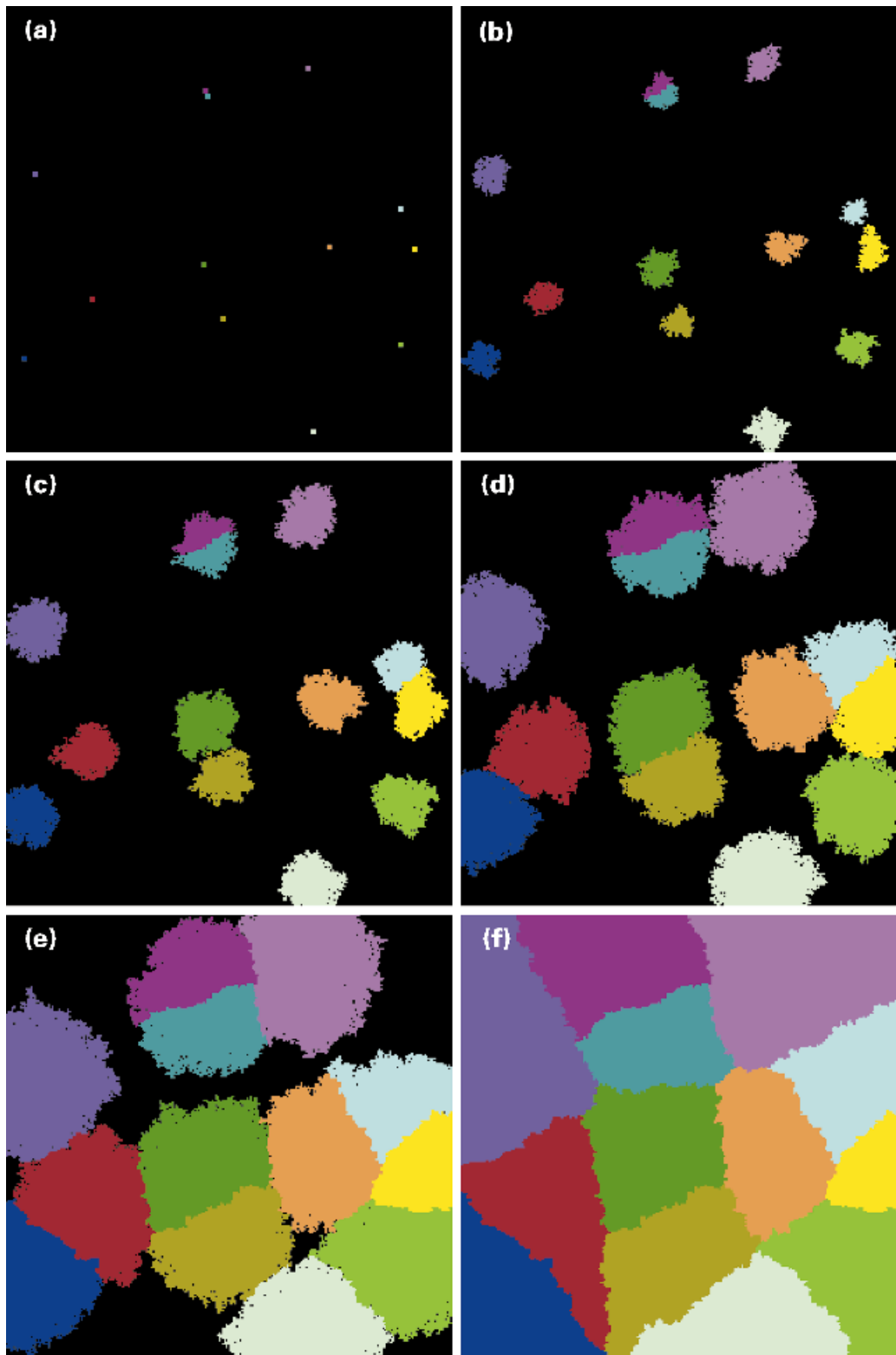
with these 'boundary grains'. One simple way to adjust for this is first to obtain a large initial tessellation and then take a middle subset of it as the final pattern. However, throughout this article, we do not perform this adjustment.

### 3. Comparison with some growth models

Eden (1961) proposed a model in order to mimic the growth process of an organism which starts with a single cell (i.e. there is only one initial pixel). Eden's growth process is the same as the growth process described here when all  $\lambda_i$ s are equal.

Williams & Bjercknes (1972) studied the growth of a tumour and suggested a model for describing the phenomenon. This model, sometimes called the contact process, has also been investigated for example by Harris (1974), Durrett (1980) and Griffeath (1981). As in Eden's model, only one initial pixel is considered. In their model, labelled pixels can be 'delabelled' and the relative probability of labelling (or delabelling) is governed by a parameter  $\kappa$ . When  $\kappa = \infty$  (i.e. labelled pixels cannot be 'delabelled'), the probability for a particular unlabelled pixel to be labelled next is proportional to the number of its labelled neighbours. This is the same as the growth process described here when the  $\lambda_i$ s are in the ratio of 1, 2, 3 and 4. On the other hand, if  $\kappa$  is very small, an object may actually shrink rather than grow.

Richardson (1973) introduced a model in which more



**Fig. 5.** A realization of the proposed growth process. (a) Initial Poisson seeds; (b) 5% of pixels are labelled; (c) 15% of pixels are labelled; (d) 40% of pixels are labelled; (e) 70% of pixels are labelled; and (f) final tessellation.

than one unlabelled pixel can be labelled in the same time step. Again, only one initial pixel is considered. In this model, each candidate for labelling (i.e. unlabelled pixels having at least one labelled neighbour) will be labelled in the next time step with probability  $p$ . So if there are  $N$  candidates for labelling, we expect  $Np$  candidates will be labelled in the next time step.

More recently, a growth process called *probabilistic mixing* has been introduced by Thompson & Rosenfeld (1993). This probabilistic mixing growth combines both deterministic and stochastic growth processes. These authors also considered the case when the initial seed is not a single pixel, but is a *skeleton*—a set of connected pixels of width 1.

A somewhat different growth process, the diffusion limited aggregation (DLA) model, is discussed for example by Witten & Sander (1981) and Meakin & Sander (1985). The first step of this DLA model is to place a seed in the centre of the image. Then a second pixel (or particle) is randomly placed in the image and is allowed to 'walk' in a random fashion. When this second particle hits the seed, it becomes part of the growing seed (grain). A third pixel is then released to walk randomly until it joins the growing seed, and then a fourth pixel is released and so on. The procedure stops when the number of pixels released hits a pre-set limit, and grains resulting from this growth process are often of 'star-shape'. Also, this DLA growth process has been extended to the case of multiple seeds; see Brémond & Jeulin (1994a) and Brémond & Jeulin (1994b).

Clearly, there is some overlap between our model and some of those above. However, most of the above models consider a single initial seed, whereas our proposed growth model is capable of handling the growth of multiple seeds and the formation of region boundaries.

## 4. Some properties of the tessellation

### 4.1. Number and effect of independent parameters

At first sight, there are six independent parameters for the proposed tessellation: the Poisson intensity  $\mu$ , an indicator parameter  $I$  for choosing the random or the majority boundary formation rule, and the four  $\lambda_i$ s. However, there are in fact only four independent parameters which have an influence on the final crystallization pattern:  $\mu$ ,  $I$ ,  $r_1 = \log(\lambda_2/\lambda_1)$  and  $r_2 = \log(\lambda_3/\lambda_1)$ . This follows, firstly because an  $x \in C_4$  has a defined fate regardless of the value of  $\lambda_4$ , and so  $\lambda_4$  is redundant, and secondly because the sum of all probabilities defined by (1) is one.

It is not hard to imagine that different combinations of  $r_1$ ,  $r_2$  and  $I$  would have different effects on the final crystallization patterns. Figure 6 attempts to demonstrate this: it seems that  $I$  tends to have a bigger impact on the local behaviour of the region boundaries (the majority rule appears to give smoother boundaries; see the next sub-

section), whereas  $r_1$  and  $r_2$  tend to affect the overall shapes of the regions.

### 4.2. Boundary roughness

As suggested before, by changing  $r_1$ ,  $r_2$  and/or applying a different boundary formation rule, we can change the boundary roughness, and hence it is desirable to have some quantitative measure of the boundary roughness. Fractal dimension is *not* suitable for the following reason. If we look at the boundaries at the microscopic level, all these boundaries consist of just horizontal and vertical straight lines. That is, they are 'smooth' and always have fractal dimension one (e.g. see Barnsley 1988) no matter how rough they appear to human eyes.

In order to investigate the effects of changing the parameter values on boundary roughness, we do the following. First, we measure the distance of the boundary of the grain from the centre of its initial pixel (or its centroid if the initial pixel is unknown) as a function of angle or arc-length. Call these distances the grain radii. It is usually convenient to choose 'equally spaced' radii. Now suppose we have chosen  $n$  radii  $l_1, \dots, l_n$  and let  $\bar{l}$  be their mean. Then we compute

$$R = \frac{\sum_i (l_{i+1} - l_i)^2}{\sum_i (l_i - \bar{l})^2}$$

as our roughness measure. Notice that  $R$  is a random function of  $r_1$ ,  $r_2$  and the boundary formation rule indicator parameter  $I$  (we assume that the Poisson intensity  $\mu$  is small enough not to have any effect on the boundary roughness). Also notice that the use of  $R$  implicitly assumes that all the grains are of 'star-shape' (of course this 'star-shape' assumption is not always satisfied, but it is not a major issue for the experimental work of this subsection; see the caption to Fig. 7).

Simulations show that  $R$  (or more precisely, the expected value of  $R$ ) heavily depends on the choice of the boundary formation rule: the random rule *always* gives a larger value of  $R$  than the majority rule with the same  $r_1$  and  $r_2$ . Also, smaller values of  $r_1$  and  $r_2$  tend to produce a larger value of  $R$  with  $r_1$  being more influential; see Fig. 7.

### 4.3. Is the tessellation isotropic?

Since we are using a four-connected system, we only have two growth directions: horizontal and vertical, and hence the stochastic growth mechanism and the final tessellation pattern are not isotropic. Somewhat surprisingly, our simulations suggest that both the growth process and the tessellation are isotropic for practical purposes, provided that the Poisson intensity is not too large. The results of formal hypothesis testing (e.g. testing differences between radii at different angles) are also consistent with a hypothesis of isotropy. Of course, this isotropic property of the tessellation pattern no longer holds if we apply the last vertical scaling step.



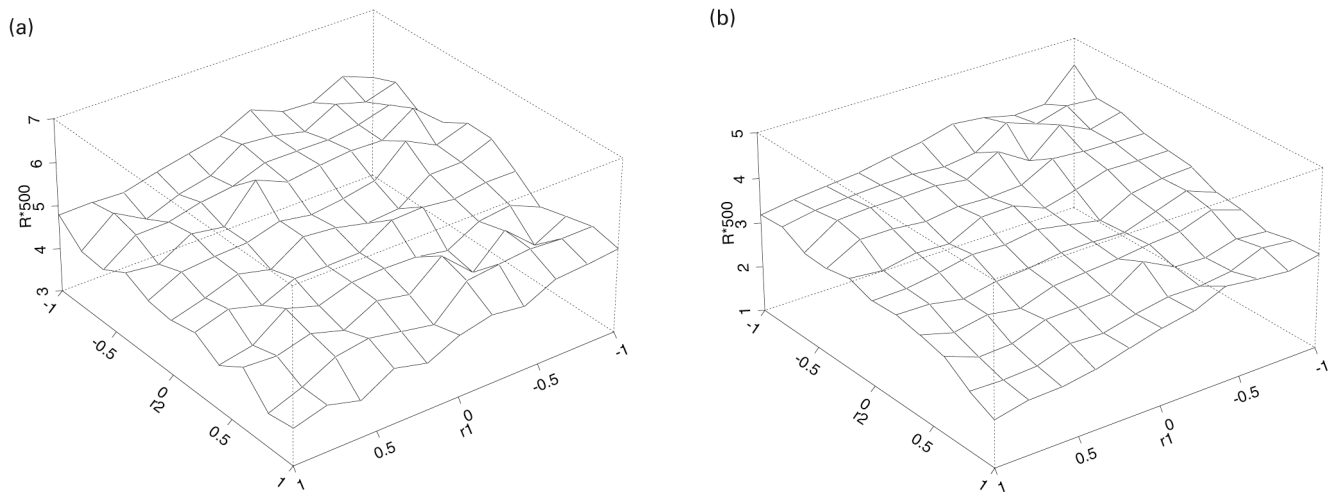
**Fig. 6.** Effects of different parameter combinations. All four realizations were started with the same set of initial seeds. (a) Random rule,  $r_1 = r_2 = 0$ ; majority rule,  $r_1 = r_2 = 0$ ; (c) random rule,  $r_1 = r_2 = 10$ ; (d) random rule,  $r_1 = r_2 = -10$ .

Figure 8 shows a small-scale simulation demonstrating the isotropic property of the growth process. Batchelor & Henry (1991) have performed a more extensive simulation for the Eden model ( $2^{25}$  pixels involved), which also demonstrates isotropy. Various researchers have attempted to obtain the asymptotic shape of different growth processes (e.g. Bramson & Griffeath, 1981; Durrett & Liggett, 1981) as the grain size increases, but the results are so far limited.

#### 4.4. Asymptotic to the Voronoi tessellation?

If the stochastic growth process is in fact isotropic, a sensible question to ask is: does the proposed tessellation converge to the Voronoi tessellation (with the same set of initial seeds) in some asymptotic sense?

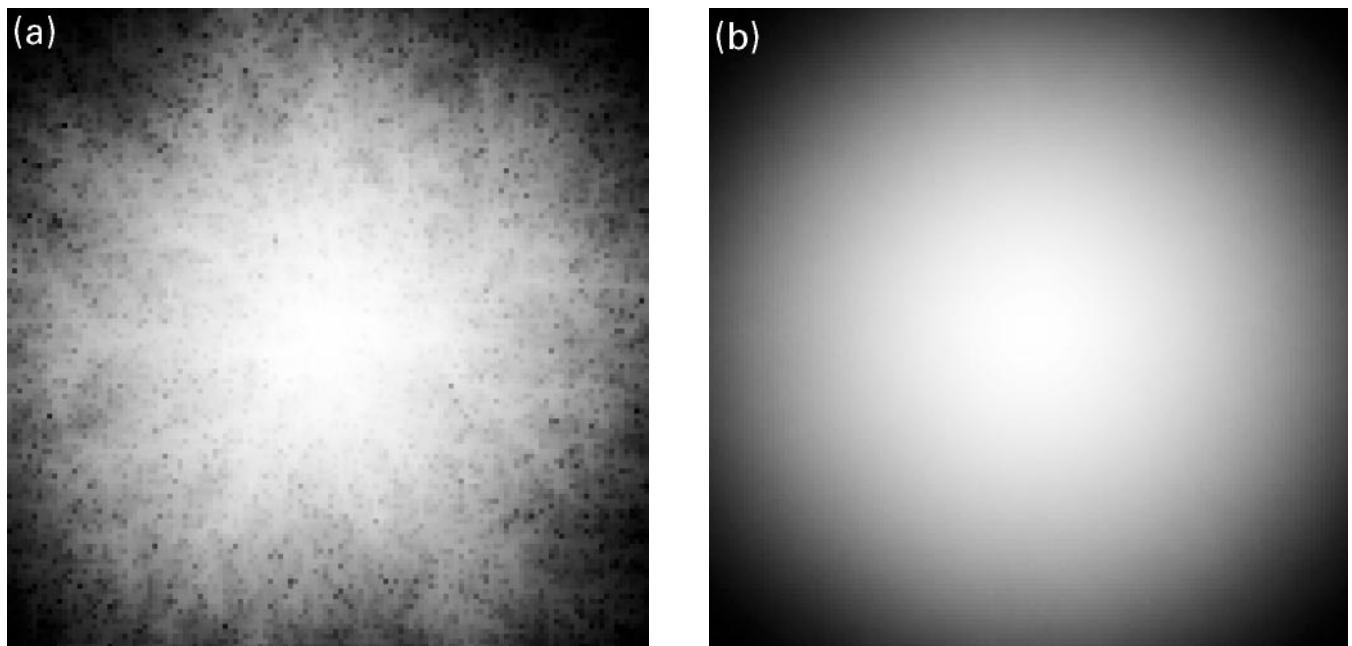
There are two common spatial asymptotics discussed in the literature (e.g. see Stein, 1995 and references therein):



**Fig. 7.** Perspective plots demonstrating the effect of various  $r_1$ ,  $r_2$  and boundary formation rules on  $R$ . Each data point is the average of 50 different simulations. It is clear that, for both boundary formation rules, the roughness measure  $R$  is a decreasing function of  $r_1$  and  $r_2$ . Another observation is: the random rule always gives a larger  $R$  than the majority rule. In order to speed up simulation times and minimize random errors, all simulations are done with the same nine regularly spaced initial pixels (since the seeds were regularly spaced, quite often the grains were all of 'star-shape'). (a) random rule; (b) majority rule.

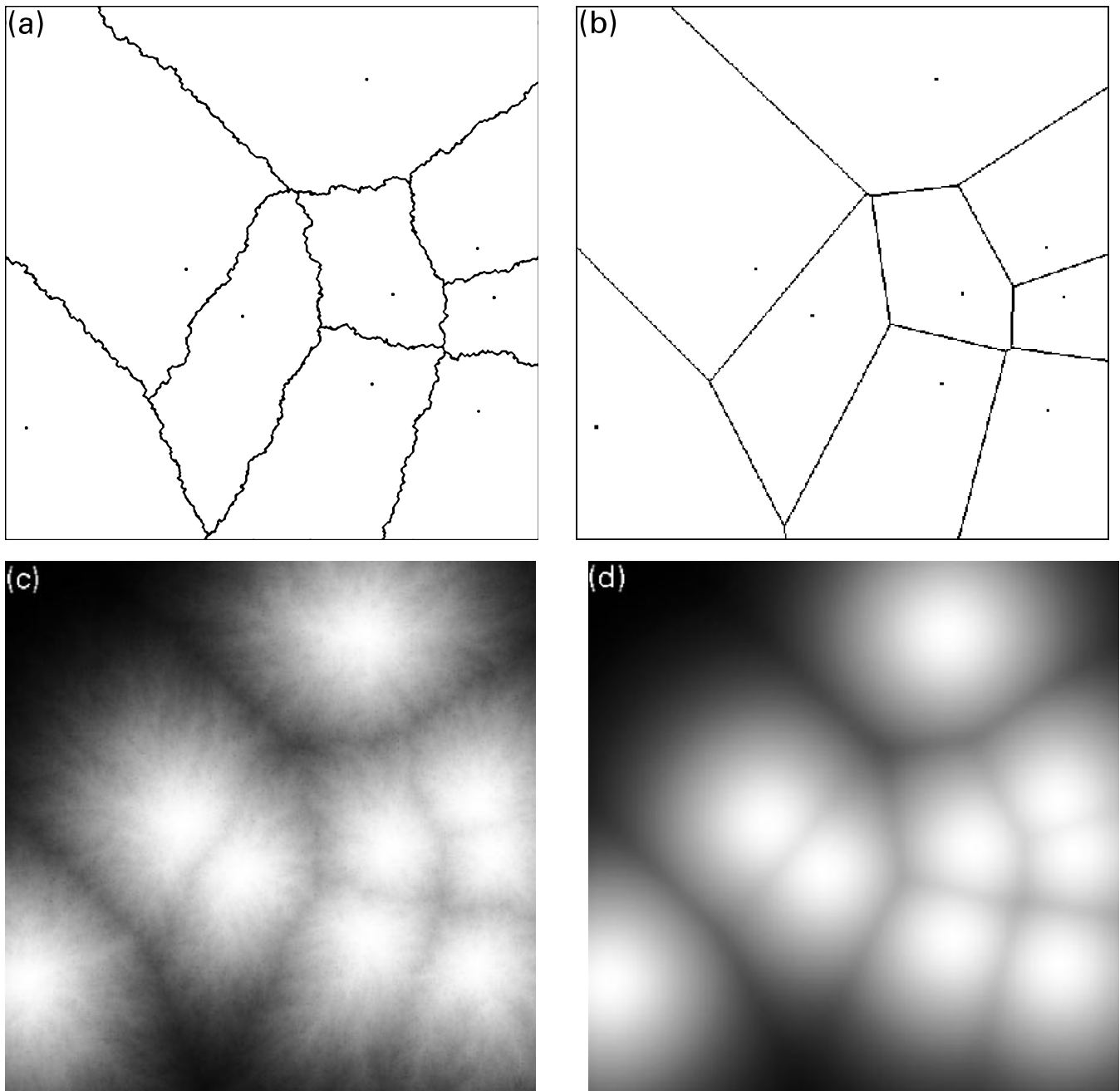
(i) increasing-domain asymptotic, in which the number of pixels in the image goes to infinity, and (ii) fixed-domain asymptotic, also called in-fill asymptotic, in which the pixel size shrinks to zero. However, we have not investigated these two asymptotics. We have, instead, briefly investigated

a simpler third property: the behaviour of the 'mean' of the boundaries of  $n$  independent realizations (all with the same seed points). Figure 9, which shows the mean of 300 such realizations, suggests that the mean boundary is a *discrete* space version of the Voronoi tessellation. We do not have any



**Fig. 8.** These two images attempt to show the time of labelling for each pixel in the domain: the brighter the pixel is, the sooner it was labelled. (a) One realization of the presented growth process (with equal  $\lambda_i$ s) with one single initial pixel located in the centre of the image; (b) average of 200 such realizations. From these images the isotropic property of the growth process is apparent. Size of such realization:  $256 \times 256$ .





**Fig. 9.** These images aim to answer the question: is the proposed tessellation asymptotic to the Voronoi tessellation? (a) One realization of the proposed tessellation together with initial Poisson seeds (black dots); (b) the corresponding Voronoi tessellation; (c) time of labelling for (a) (see caption of Fig. 8 for an explanation); (d) averages of 300 realizations of time of labelling.

theoretical results supporting our claim, only empirical results such as that shown in Fig. 9.

### 5. Possible variants

Variants of the proposed tessellation model are of course possible. For example, we can start the growth process with

non-Poisson initial seeds, or we can modify the growth process such that the growth in the horizontal (or the vertical) direction is faster than the growth in the other direction (this issue will be further discussed in Section 6). Below we discuss two simple but interesting variants: a Johnson–Mehl type modification and a three-dimensional (3D) extension.

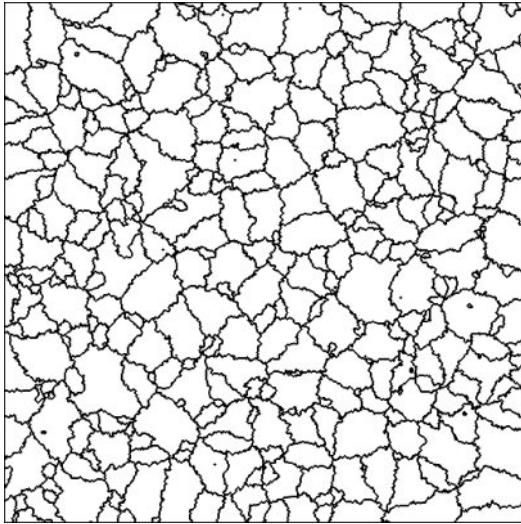


Fig. 10. A realization of the Johnson–Mehl type modification.

### 5.1. Johnson–Mehl type modification

A simple modification to the proposed tessellation is to allow the birth of new seeds at any time step during the growth process. This kind of modification is often called a Johnson–Mehl type modification (e.g. see Stoyan *et al.*, 1987, Chapter 10). It often produces more interesting and complex patterns than the original model. For example, it is easy to find a smaller grain lying totally inside a bigger grain. See Fig. 10 for a realization.

### 5.2. Three-dimensional extension

The formulation of the proposed model permits simple and straightforward higher dimensional extensions. Figure 11 displays different cross-sections of a realization of such a 3D extension. One noteworthy observation is that if a (3D) grain has a U-shape, then in some cross sections it may appear as two distinct grains.

## 6. Shape modelling of aluminium grain images

This section demonstrates how the proposed tessellation model can be applied to simulate the shape of the aluminium grains. The semi-automatically segmented aluminium grain image displayed in Fig. 2 will be taken as the target for our simulation.

### 6.1. Choice of model parameters

Recall that there are four independent model parameters

(see Section 4.1):  $\lambda$ ,  $I$ ,  $r_1$  and  $r_2$ . Given a segmented image such as the one displayed in Fig. 2, the intensity of the Poisson seeds  $\lambda$  can be estimated by

$$\frac{\text{number of segmented grains}}{\text{area of the image}}.$$

(This estimator for  $\lambda$  is biased as it suffers from edge effects, but since this is not an important issue here it is ignored.)

Our empirical experience suggests that the effects of  $I$ ,  $r_1$  and  $r_2$  are somewhat confounded. This makes estimation of  $I$ ,  $r_1$  and  $r_2$  difficult. So we arbitrarily choose

$$I = \text{'random boundary formation rule'}$$

and

$$r_1 = r_2 = 1$$

to simulate the shape pattern.

### 6.2. Choice of scaling factor

As the widths of the aluminium grains tend to be longer than the heights, a further step of rescaling with respect to the vertical direction is necessary. Two simple statistics,  $\rho_{\text{MEAN}}$  and  $\rho_{\text{MAX}}$ , has been used to estimate the scaling factor of the image displayed in Fig. 2. They are defined as

$$\rho_{\text{MEAN}} = \frac{1}{k} \sum_{j=1}^k \frac{\text{width}_{\text{MEAN},j}}{\text{height}_{\text{MEAN},j}}$$

and

$$\rho_{\text{MAX}} = \frac{1}{k} \sum_{j=1}^k \frac{\text{width}_{\text{MAX},j}}{\text{height}_{\text{MAX},j}},$$

where

$$\text{width}_{\text{MEAN},j} = (\text{area of the } j\text{th segmented grain}) / (\text{number of rows it occupies}),$$

$$\text{height}_{\text{MEAN},j} = (\text{area of the } j\text{th segmented grain}) / (\text{number of columns it occupies}),$$

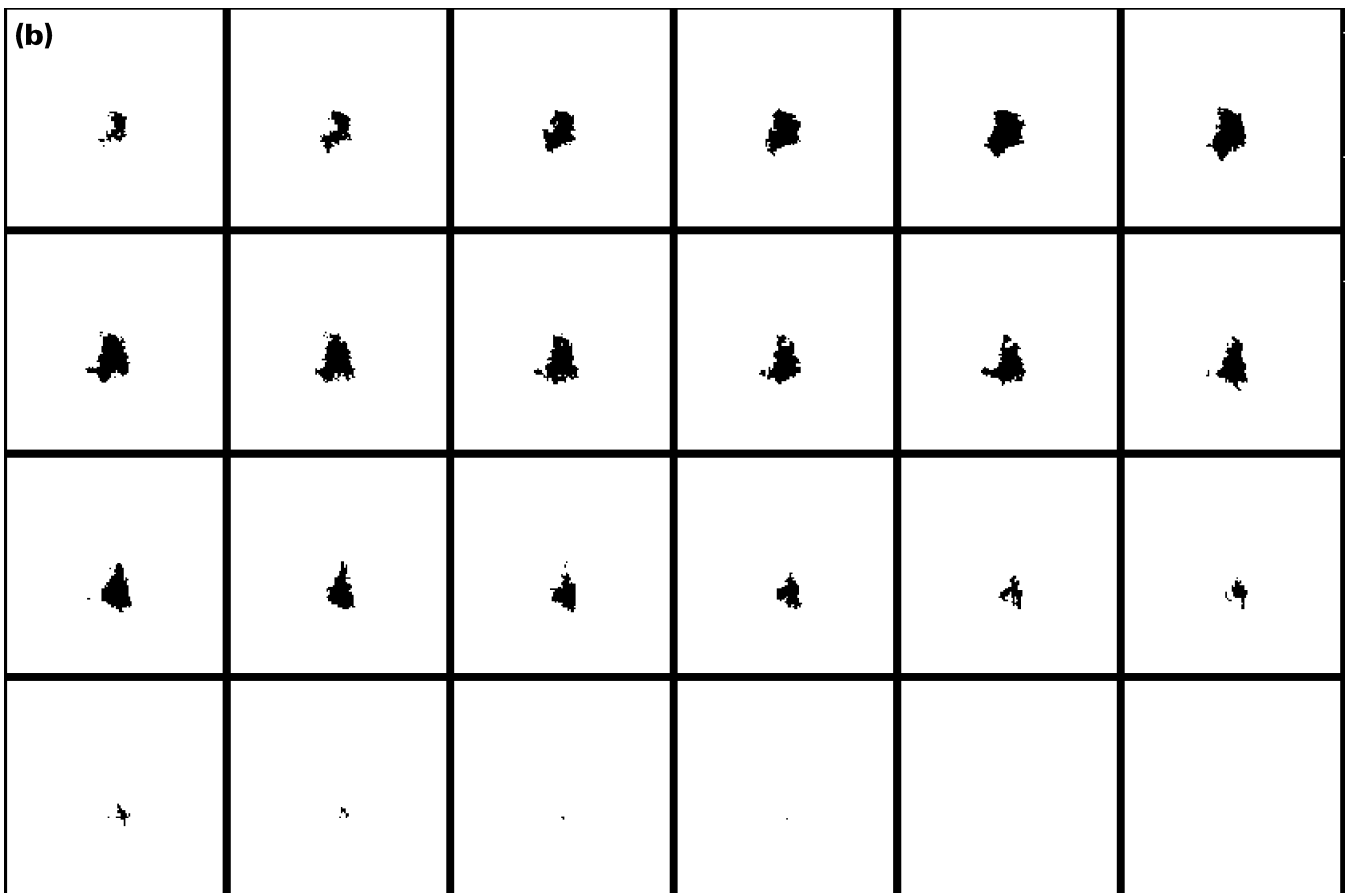
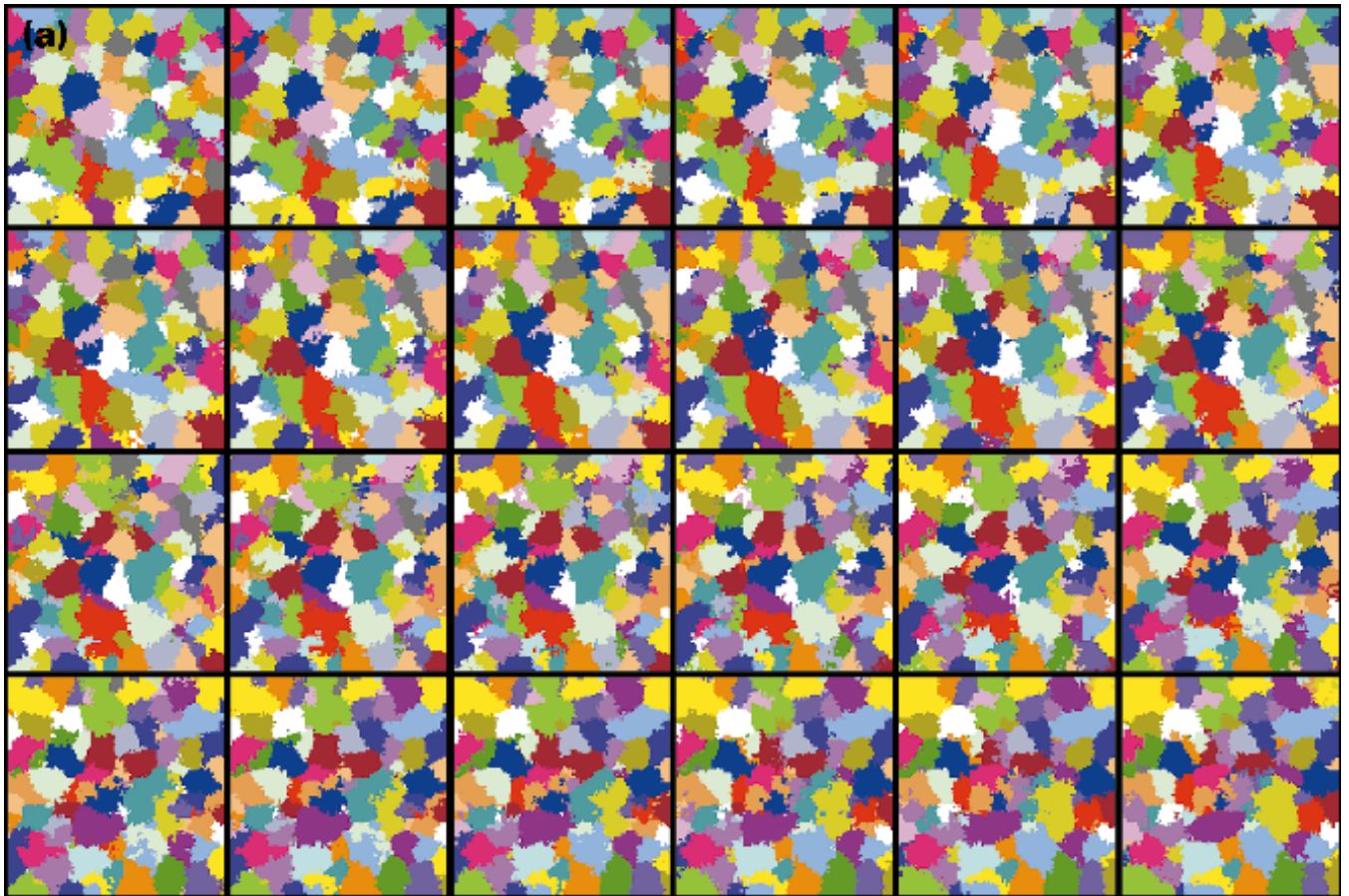
$$\text{width}_{\text{MAX},j} = \text{maximum number of pixels the } j\text{th segmented grain occupies in one row},$$

$$\text{height}_{\text{MAX},j} = \text{maximum number of pixels the } j\text{th segmented grain occupies in one column},$$

and  $k$  is the total number of segmented grains in the image. For the segmented image in Fig. 2, both  $\rho_{\text{MEAN}}$  and  $\rho_{\text{MAX}}$  produced estimates remarkably close to 2, and so we took the scaling factor to be 2. (See Fig. 19(b) for such a scaled tessellation.)

Other than rescaling, another way to generate elongated grains is to modify the growth process in such a way that

Fig. 11. Three-dimensional extension. (a) 24 successive cross-sections of one  $128 \times 128 \times 128$  realization. (b) one single highlighted grain.



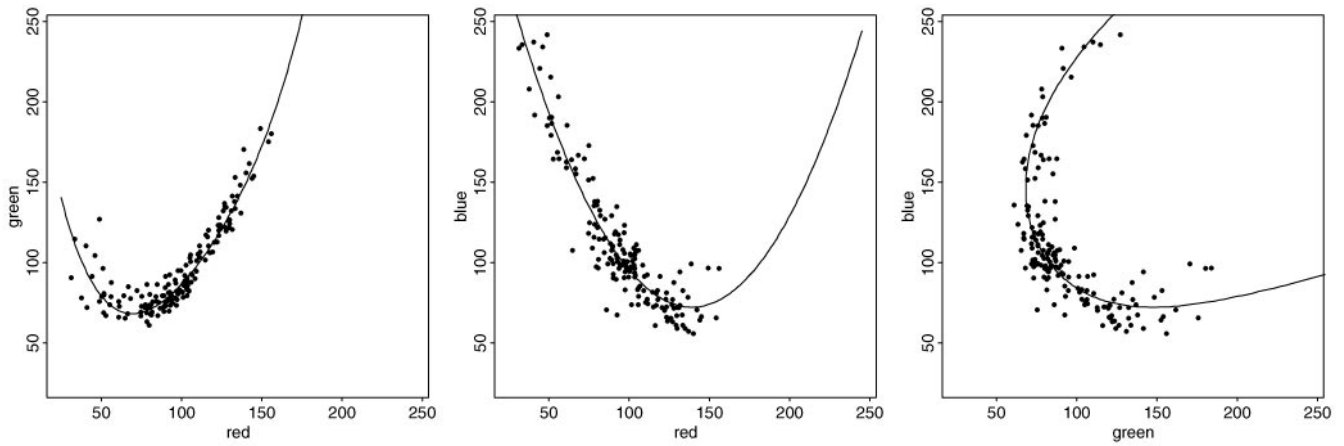


Fig. 12. Pairwise plots of raw RGB values, with fitted model (2) in solid lines.

the growth rate in the horizontal direction is faster than the vertical direction. However, we did not pursue this approach. The reasons are that it is unclear how the relative growth rate can be estimated from an image and that the real aluminium grains were in fact flattened (rescaled) after their formation.

## 7. Colour modelling

The primary concern of this section is the modelling of the *colour* characteristics of the aluminium grain images. Once a scaled tessellation is obtained (e.g. by using the tessellation model described before), the next stage for simulating an aluminium grain image is to assign a colour value to each of the simulated 'empty' regions. We first note that, for a given real aluminium grain image, it is not always possible to tell whether a region with all its pixels having similar colour values is in fact one true grain, or is composed of two or more neighbouring grains with similar colours. In these situations, we treat such a region as a single grain, and we will ignore the spatial relationship amongst the colour values when assigning simulated colours to the empty regions. We shall concentrate on the modelling of the distribution of the colour values.

### 7.1. A quadratic model for colour values

In digital colour images, a particular colour is stored as three values: the red, green and blue (RGB) components. Therefore in our modelling, colours are treated as realizations of trivariate random variables. Given a real aluminium grain image, how can one extract data in a format that can be worked with (i.e. a series of trivariate colour values)? One way to achieve this is to first segment the image, and then, for each segmented grain, compute the mean RGB values. These computed mean values, denoted by  $(r_j, g_j, b_j)$ ,  $j = 1, \dots, n$ , with  $n$  as the total number of RGB values, can be taken as

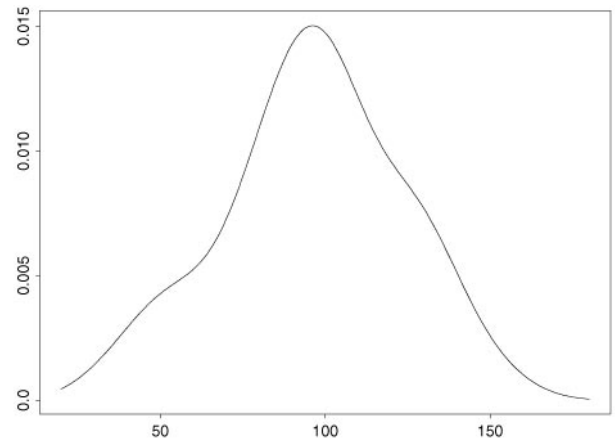


Fig. 13. Nonparametric density estimate of the  $r_j$ s.

the 'workable data'. The image in Fig. 2 has 164 segmented grains, and the corresponding  $(r_j, g_j, b_j)$ s are plotted pairwise in Fig. 12. These extracted RGB values will be taken as the observed data that we will work with. Figure 12 seems to suggest that these observed RGB values are related by a quadratic relationship in three-dimensional (3D) colour space.

Let  $(R_j, G_j, B_j)$  be the 'true' but unobservable colour of the  $i$ th segmented grain. We propose using the following simple but effective model for the observed RGB values  $(r_j, g_j, b_j)$ s:

$$\begin{aligned} r_j &= R_j \sim N(\mu_R, \sigma_R^2), \\ b_j &= B_j + \epsilon_j = \alpha_0 + \alpha_1 R_j + \alpha_2 R_j^2 + \epsilon_j \\ g_j &= G_j + \delta_j = \beta_0 + \beta_1 R_j + \beta_2 B_j + \beta_3 R_j^2 \\ &\quad + \beta_4 B_j^2 + \beta_5 R_j B_j + \delta_j \end{aligned} \quad (2)$$

where the  $\alpha$ s and  $\beta$ s are model parameters, and  $R_j$ ,  $\delta_j$  and  $\epsilon_j$  are independent Gaussian random variables:  $\delta_j \sim N(0, \sigma_\delta^2)$ ,

Table 1. Listing of parameter estimates.

Parameters	$\mu_R$	$\sigma_R^2$	$\alpha_0$	$\alpha_1$	$\alpha_2$	$\sigma_\epsilon^2$	$\beta_0$	$\beta_1$	$\beta_2$	$\beta_3$	$\beta_4$	$\beta_5$	$\sigma_\delta^2$
Estimates	96.40	26.63 <sup>2</sup>	367.8	-4.253	0.01530	14.43 <sup>2</sup>	292.4	-3.177	-1.956	0.01667	0.005277	0.008784	6.882 <sup>2</sup>

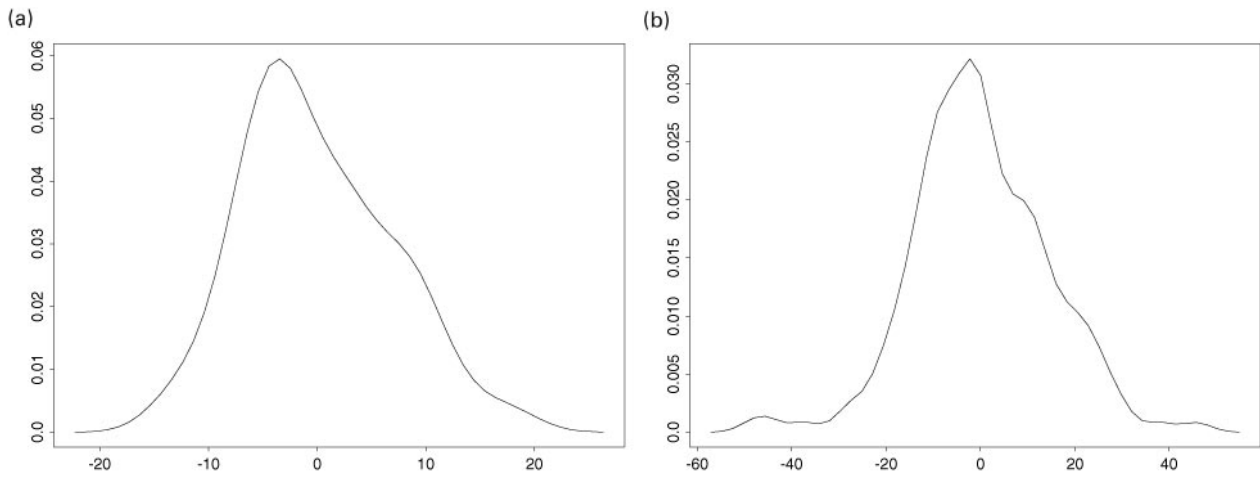


Fig. 14. Nonparametric density estimates of residuals of  $g_j$  (a) and  $b_j$  (b).

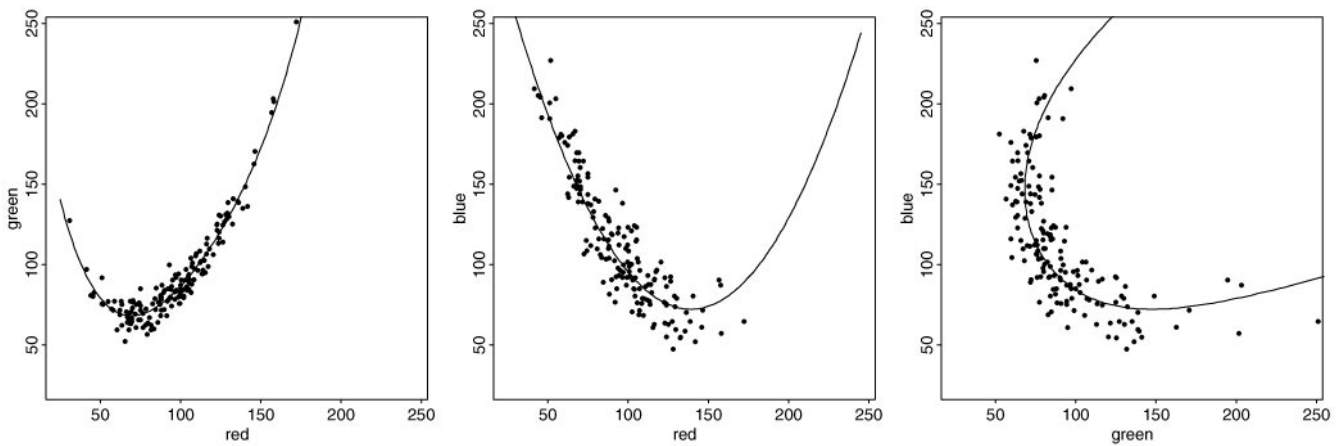


Fig. 15. Pairwise plots of simulated colour values. Also plotted in solid lines is the fitted model (2). Compare these plots with Fig. 12.

$\epsilon_j \sim N(0, \sigma_\epsilon^2)$  (by  $X \sim N(\mu, \sigma^2)$  we mean  $X$  is a Gaussian random variable with mean  $\mu$  and variance  $\sigma^2$ ). Notice that in the above model both  $g_j$  and  $b_j$  are assumed to be observed with errors ( $\delta_j$  and  $\epsilon_j$ ), whereas  $r_j$  is error-free, and is assumed to be Gaussian. Figure 13 is a nonparametric density estimate (e.g. see Silverman, 1986) of  $r_j$ , which empirically justifies the assumed Gaussianity of the  $r_j$ s. A more sophisticated model in which  $r_j$ ,  $g_j$  and  $b_j$  are all assumed to be error-contaminated can be found in Lee (1997).

We understand that the above quadratic model Eq. (2) does not give any physical explanation of the generation of the observed ‘true’ colour values. However, because

of its simplicity and efficiency, we recommend its use here.

We computed the maximum likelihood estimates (which are also the least-square estimates in this situation) of the parameters. They are listed in Table 1. The corresponding fitted 3D quadratic curve is plotted in Fig. 12. Also, nonparametric density estimates of the residuals of the fitted  $g_j$  and  $b_j$  are plotted in Fig. 14; these residuals are reasonably zero-mean Gaussians.

### 7.2. Simulation of artificial aluminium grain colours

To simulate an artificial aluminium grain colour value, we

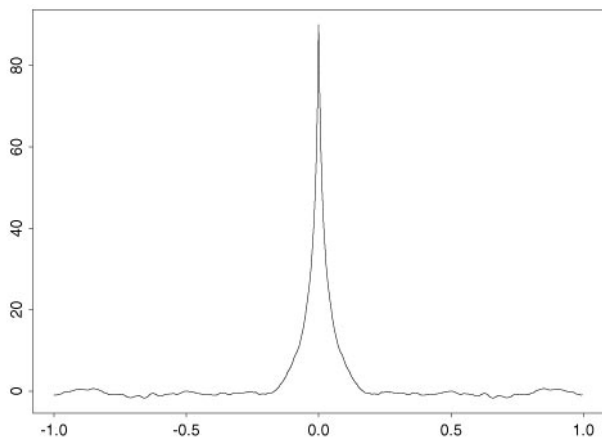


Fig. 16. Cross-section of the empirical autocovariance of extracted noise.

first simulate an  $r_j$  from  $N(\hat{\mu}_R, \hat{\sigma}_R^2)$ , where  $\hat{\mu}_R = 96.40$  and  $\hat{\sigma}_R^2 = 26.63^2$ , and then simulate  $g_j$  and  $b_j$  using model (2) with parameter estimates given in Table 1. Figure 15 displays pairwise plots of such a set of simulated colour values, and these simulated colour values can be assigned to the empty regions of a tessellation to form an artificial (noise-free) colour aluminium grain image. However, one additional step is required after each empty region is assigned with a colour: if the Euclidean distance between the simulated colour values of two neighbouring regions is less than a pre-set threshold, these two neighbouring regions are merged together. Because of this additional

merging step, the intensity of the Poisson seeds for simulating a tessellation should be increased. (Figure 19(c) provides an example of a simulated colour image.)

## 8. Noise modelling

This section considers the modelling of the *noise* characteristics of the aluminium grain images. A careful examination of the aluminium grain images suggests that the noise is *spatially correlated*; Fig. 16 displays a cross-section of the empirical autocovariance function of the 'extracted noise' (see below) of the red band, which supports the claim that the noise is spatially correlated. It will be convenient to model this noise using a stationary Gaussian random field (SGRF), and hence it has been decided to use SGRFs to model the noise (e.g. see Chellappa & Kashyap, 1985). Therefore our aim is: given an observed noise image which can be well modelled by an SGRF  $T$ , we would like to simulate other noise images (i.e. other SGRFs) which would *look like* and *possess similar statistical properties* to  $T$ .

We shall use the technique developed in Lee & Berman (1997) to achieve our aim. Specifically, Lee & Berman (1997) proposed an automatic spectrum estimation procedure, which is designed to produce an estimated spectrum of an observed SGRF  $T$  in such a way that any subsequent SGRFs simulated from this estimated spectrum would look like and possess similar statistical properties to  $T$ .

Before we proceed, we first mention one legitimate suspicion, which is that the noise is not only spatially correlated, but is also correlated *across* the colour bands

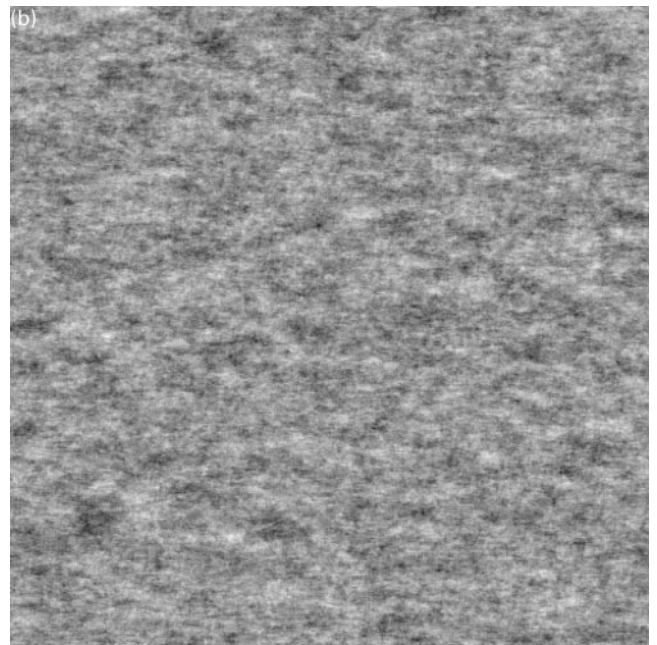
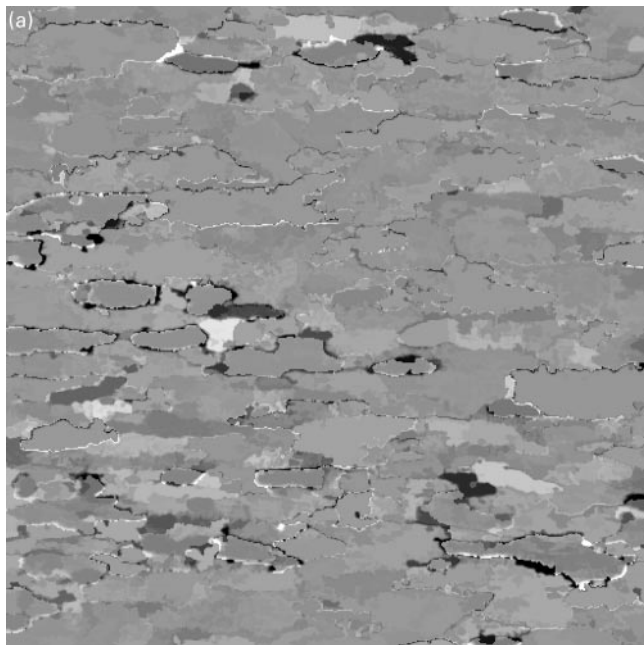


Fig. 17. Extracted noise (a) and the corresponding simulated noise (b).

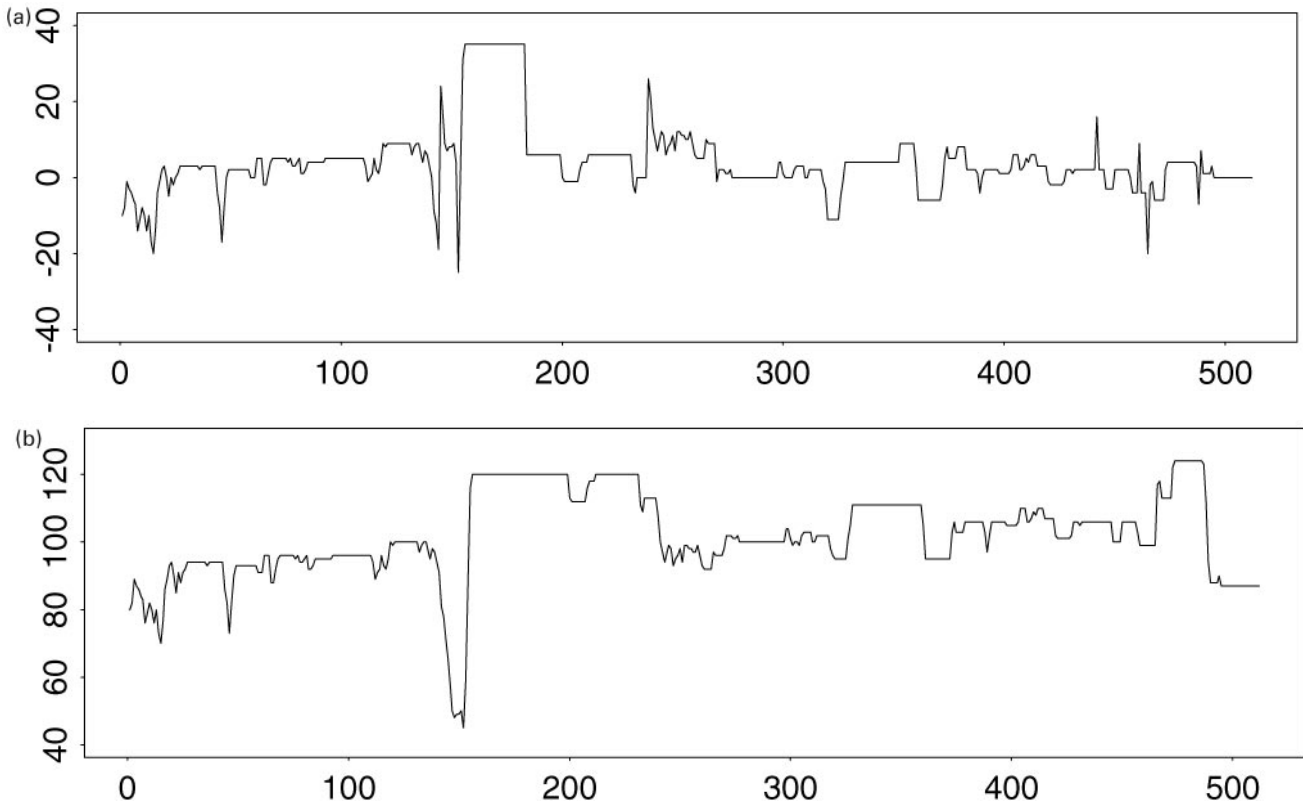


Fig. 18. Cross-sections of the extracted noise (a) and the corresponding red band of the aluminium grain image (b). The square feature of some parts of the plots is due to a discretization effect when recording the image.

(e.g. the noise of the red band is correlated with the noise of the green band). However, for simplicity, this across-band correlation will be ignored.

### 8.1. Noise modelling of aluminium grain images

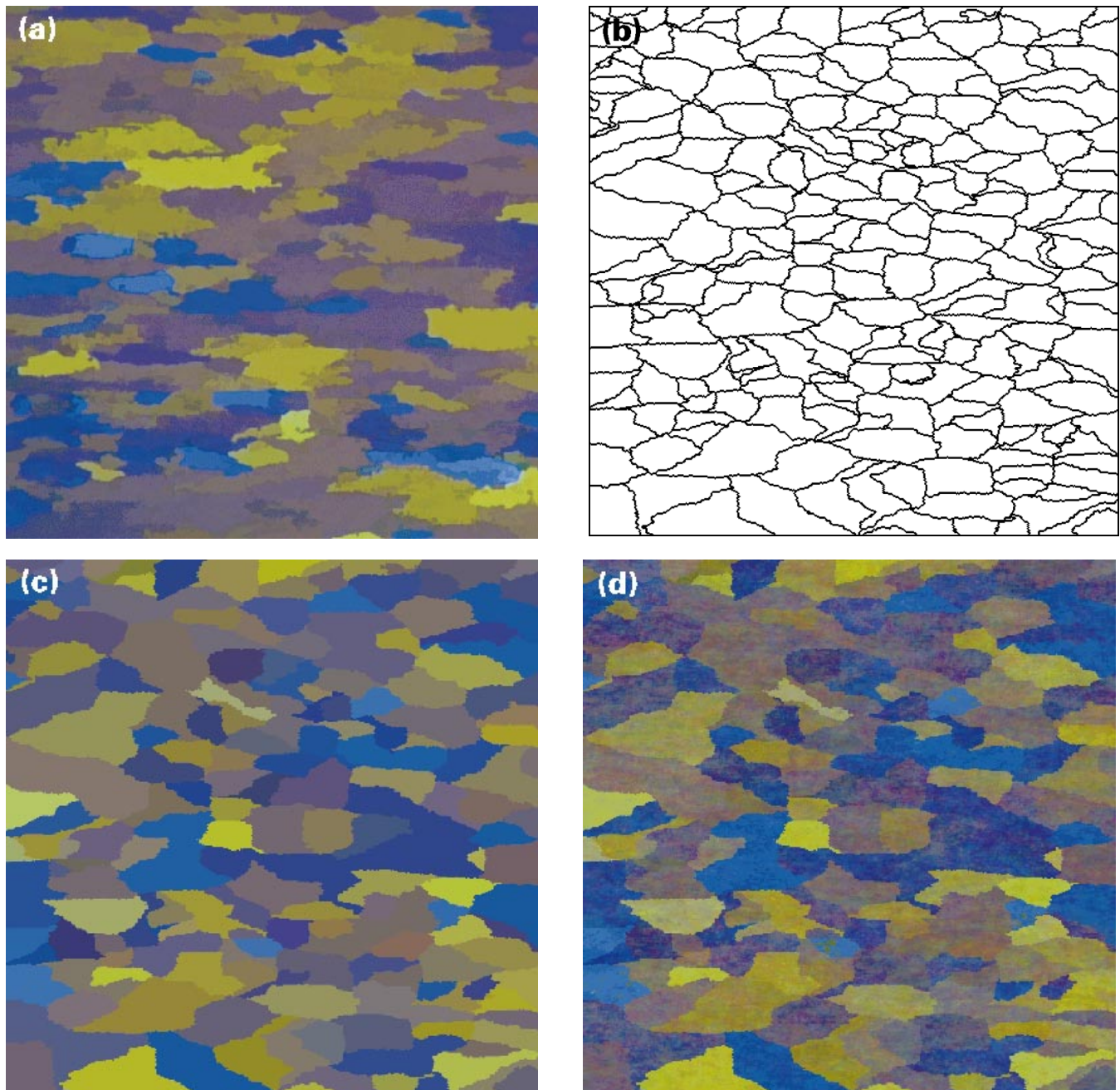
As for the colour modelling, given a real aluminium grain image, the first step of our noise modelling is to extract data in a workable format. We can apply the same trick: given a real image, we segment it and compute the mean colour values for each segmented grain. Then the difference between this mean-computed segmented image and the real observed image can be taken as the noise. See Fig. 17(a) for the 'extracted noise' of the red band of the image in Fig. 2.

Observe that there are some strong, narrow and long features around the grain boundaries, and that the extracted noise may not be well modelled by an SGRF. Even though these features are present, we applied the technique developed by Lee & Berman (1997) to estimate the spectrum of the noise in Fig. 17(a), and from the estimated spectrum, an SGRF is generated; see Fig. 17(b).

When comparing visually the simulated noise in Fig. 17(b) with the extracted noise in Fig. 17(a), however, the result is unsatisfactory. We attribute this failure partially to the fact that the extracted noise cannot be well modelled by an SGRF. In Fig. 18 is the cross-section of the horizontal middle line of the extracted noise, which clearly shows that there are sharp jumps in the extracted noise. Such jumps, which are artifacts resulting from the imperfection of the semi-automatic segmentation and the blurred grain boundaries, violate the smoothness requirement of SGRFs, which has, in turn, caused the above failure. Nevertheless, we ignored this failure and added one such simulated SGRF (from the estimated spectrum of the extracted noise) to each of the three components of a simulated colour image to form our fully simulated aluminium grain image; see Fig. 19(d).

## 9. Comparing real and simulated images

A visual comparison of a real aluminium grain image, Fig. 19(a), and a final simulated image, Fig. 19(d), suggests that our model generates tessellations which have more regular boundaries than do the real aluminium grains. This can be partially overcome by a 3D version of the tessellation model



**Fig. 19.** Modelling of aluminium grain image. (a) Real aluminium grain image; (b) shape modelling; (c) shape and colour modelling; (d) shape and colour and noise modelling.

(see Section 5.2). Because the 3D simulations take much longer, we recommend using the 2D version.

In order to compare the real and the simulated images in a quantitative fashion, we computed nonparametric density estimates of the area and the perimeter of the grains in the real and simulated images. These estimates are plotted in Figs 20(a) and (b). Also, we fitted an ellipse to each individual grain. Density estimates of the ratios of the length of the fitted ellipse major axes to the length of the minor

axes are plotted in Fig. 20(c), while estimates of the orientation of the major axes are plotted in Fig. 20(d). From these plots we can see that the real and simulated images are similar in global characteristics.

## 10. Concluding remarks

In this article we have described various models and methods for simulating artificial colour aluminium grain



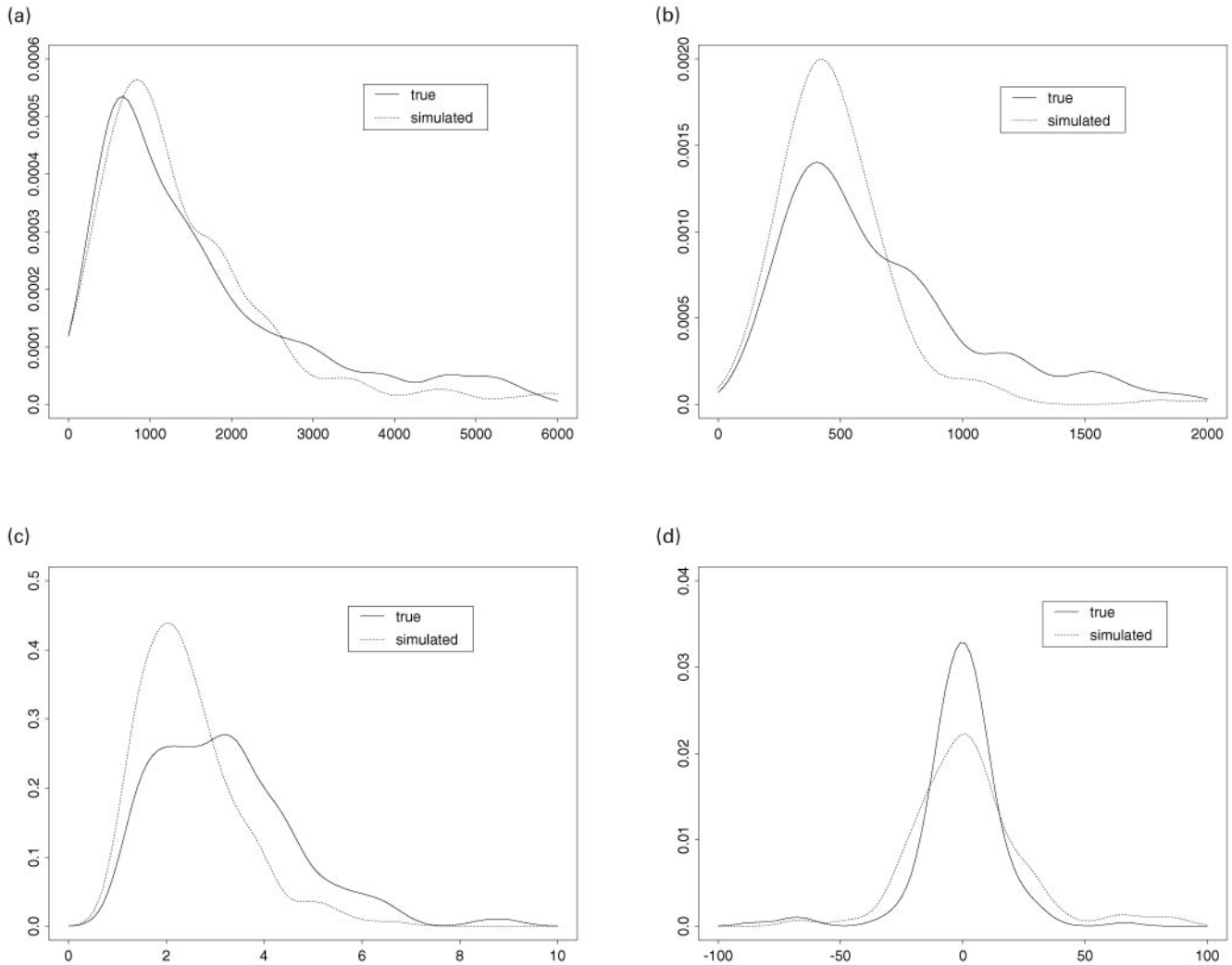


Fig. 20. Comparing real and simulated images. (a) area; (b) perimeter; (c) axes ratio; (d) orientation.

images. In particular, a new tessellation model is proposed. This tessellation model serves our purposes well: it is capable of generating tessellations with rough boundaries, which can be used as the basis of building synthesized aluminium grain images. In addition, it does not need a long computation time to obtain one realization: for an image of dimension  $512 \times 512$  with about 220 initial seeds, it takes about 150 s of user time on a Sparc-10 workstation.

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